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Key indicators

Single-crystal X-ray study $T=295~\mathrm{K}$ Mean $\sigma(\mathrm{Si-Si})=0.004~\mathrm{Å}$ R factor = 0.022 wR factor = 0.050 Data-to-parameter ratio = 12.1

For details of how these key indicators were automatically derived from the article, see http://journals.iucr.org/e.

Erbium dicobalt disilicide, ErCo₂Si₂

Single crystals of erbium dicobalt disilicide were synthesized from the corresponding elements by arc melting. The ternary intermetallic compound crystallizes in the body-centred tetragonal space group *I4/mmm* and adopts the CeGa₂Al₂ structure type, with all atoms in special positions.

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Comment

Ternary intermetallics of rare earth metals with the general formula RET_2X_2 (where RE is a rare earth metal, T is a transition metal or, in a few cases, a p-block element, and X is a p-block element) crystallize mostly in ten structure types: tetragonal $CeGa_2Al_2$ (also called $ThCr_2Si_2$), $CaBe_2Ge_2$ and YB_2C_2 , trigonal La_2O_2S , orthorhombic $CaRh_2B_2$, $LaRe_2Si_2$, $HfFe_2Si_2$ and ScB_2C_2 , and monoclinic $LaPt_2Ge_2$ and $HoNi_2B_2$. In the majority of cases, RET_2X_2 phases belong to the $CeGa_2Al_2$ structure type (body-centred tetragonal) or to the very similar $CaBe_2Ge_2$ structure type (primitive tetragonal), which are both ordered derivatives of the $BaAl_4$ structure (Parthé et al, 1983).

The RET_2X_2 phases have received special attention due to their interesting physical properties. The compound $CeCu_2Si_2$ ($CeGa_2Al_2$ type) was the first representative of a heavy-fermion system. The accurate determination of the crystal structure for phases of this composition is necessary for a better understanding of their physical properties. The existence of the $ErCo_2Si_2$ phase was first reported by Rossi *et al.* (1978), and the crystal structure was determined by means of X-ray powder diffraction measurements. Subsequently,

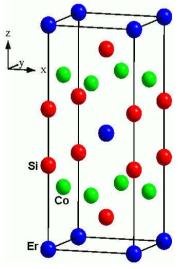


Figure 1 A clinographic projection of the ErCo₂Si₂ unit cell, with displacement ellipsoids drawn at the 95% probability level.

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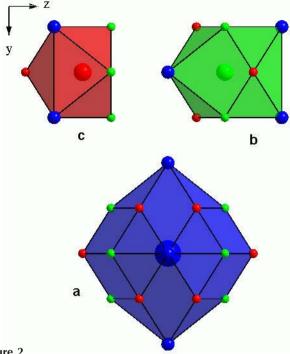


Figure 2
Coordination polyhedra around (a) the Er atom, (b) the Co atom and (c) the Si atom. Er atoms are blue, Co atoms are red and Si atoms are green.

neutron powder diffraction measurements and structure refinements have been performed by Yakinthos *et al.* (1983) and Leciejewicz *et al.* (1983). In view of the close relation of the structure types CeGa₂Al₂ (space group *I4/mmm*) and CaBe₂Ge₂ (space group *P4/mmm*), it was necessary to determine precisely the structure type for ErCo₂Si₂ on the basis of single-crystal diffraction data, and we present these results here.

 ${\rm ErCo_2Si_2}$ adopts the ${\rm CeGa_2Al_2}$ structure type. A clinographic projection of the unit cell is shown in Fig. 1. The coordination sphere around Er (site symmetry 4/mmm) consists of 22 atoms, resulting in a polyhedron with 22 vertices ${\rm [ErSi_8Co_8Si_2Er_4]}$ (Fig. 2a), and 12 quadrangular and 24 triangular faces. The coordination polyhedron of the Co atom (site symmetry $\overline{4}m2$) is a distorted cuboctahedron ${\rm [CoSi_4-Co_4Er_4]}$ (Fig. 2b). The coordination polyhedron for Si (site symmetry 4mm) is a monocapped tetragonal antiprism ${\rm [SiCo_4Si_1Er_4]}$ with one additional Si as the capping atom (Fig. 2c). The interatomic distances are in good agreement with the sums of the atomic radii (Pauling, 1967). The shortest distance (Table 1) is observed between Co and Si atoms (94% of the sum of the atomic radii of the corresponding atoms).

Experimental

The single crystal used in this work was extracted from an alloy with nominal composition $\rm Er_{20}Co_{40}Si_{40}$, which was prepared by arc melting of the initial components (purity better than 99.9%) in an electric arc furnace with a water-cooled copper bottom (Ti-getter) under an argon atmosphere and annealed at 870 K. A preliminary crystal investigation was performed using Laue and rotation methods (RKV-86 and RGNS-2 chambers, Mo $K\alpha$ radiation).

Crystal data

Co₂ErSi₂ Mo $K\alpha$ radiation $M_r = 341.30$ Cell parameters from 561 Tetragonal, I4/mmm reflections a = 3.874 (2) Å $\theta = 2.4-29.3^{\circ}$ $\mu = 40.29 \text{ mm}^{-1}$ c = 9.704 (4) Å $V = 145.64 (12) \text{ Å}^3$ T = 295 (2) KZ = 2Plate, metallic light grey $D_x = 7.783 \text{ Mg m}^{-3}$ $0.14 \times 0.13 \times 0.04 \text{ mm}$

Data collection

Oxford Xcalibur3 CCD areadetector diffractometer a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ and $I > 2\sigma(I)$ and $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consistence with $I > 2\sigma(I)$ and $I > 2\sigma(I)$ are a consiste

Refinement

 $\begin{array}{lll} \text{Refinement on } F^2 & (\Delta/\sigma)_{\text{max}} < 0.001 \\ R[F^2 > 2\sigma(F^2)] = 0.022 & \Delta\rho_{\text{max}} = 2.09 \text{ e Å}^{-3} \\ wR(F^2) = 0.050 & \Delta\rho_{\text{min}} = -2.41 \text{ e Å}^{-3} \\ S = 1.17 & \text{Extinction correction: } SHELXL97 \\ 109 \text{ reflections} & (\text{Sheldrick, 1997}) \\ 9 \text{ parameters} & w = 1/[\sigma^2(F_o^2) + (0.0245P)^2 \\ & + 2.3748P] \\ \text{where } P = (F_o^2 + 2F_c^2)/3 \end{array}$

Table 1 Selected geometric parameters (Å, °).

Er-Si ⁱ	2.9977 (17)	Co-Co ⁱⁱⁱ	2.7393 (14)
Er-Co ⁱ	3.1044 (10)	Si-Si ^{iv}	2.435 (6)
Co-Si ⁱⁱ	2.2831 (17)		
Si^{i} - Er - Si^{v}	180.0	Si ^x -Co-Co ⁱⁱⁱ	53.14 (3)
Si^{i} - Er - Si^{vi}	132.07 (10)	Co ⁱⁱⁱ -Co-Co ⁱⁱ	180.0
Si ^v -Er-Si ^{vi}	47.93 (10)	Co^{iii} $-Co$ $-Co^{xi}$	90.0
Si^{i} - Er - Si^{vii}	80.51 (4)	Si ⁱⁱ -Co-Er	65.57 (5)
Si^v - Er - Si^{vii}	99.49 (4)	Si^{x} -Co-Er	160.56 (6)
Si ⁱ -Er-Co ⁱ	43.90 (4)	Si-Co-Er	83.35 (7)
Si ^v -Er-Co ⁱ	136.10 (4)	Co ⁱⁱ -Si-Co ^{xii}	73.73 (6)
Si ^{vi} -Er-Co ⁱ	94.92 (5)	Co ⁱⁱ -Si-Co ^v	116.08 (13)
Si ^{viii} -Er-Co ⁱ	85.08 (5)	Co ⁱⁱ -Si-Si ^{iv}	121.96 (6)
Co ⁱ -Er-Co ^{ix}	52.36 (2)	Co ⁱⁱ -Si-Er ^{xiii}	139.747 (19)
Si ⁱⁱ -Co-Si ^x	106.27 (6)	Co ^v -Si-Er ^{xiii}	70.54(2)
Si ⁱⁱ -Co-Si ^v	116.08 (13)	Si^{iv} - Si - Er^{xiii}	66.04 (5)
Si ⁱⁱ -Co-Co ⁱⁱⁱ	126.86 (3)	Er ^{xiii} —Si—Er ^{xiv}	132.07 (10)

Symmetry codes: (i) $x-\frac{1}{2}$, $y-\frac{1}{2}$, $z-\frac{1}{2}$; (ii) $-\frac{1}{2}-x$, $\frac{1}{2}-y$, $\frac{1}{2}-z$; (iii) $\frac{1}{2}-x$, $\frac{3}{2}-y$, $\frac{1}{2}-z$; (iv) -x, -y, 1-z; (v) $\frac{1}{2}-x$, $\frac{1}{2}-y$, $\frac{1}{2}-z$; (vi) $\frac{1}{2}+x$, $\frac{1}{2}+y$, $z-\frac{1}{2}$; (vii) $x-\frac{1}{2}$, $\frac{1}{2}+y$, $z-\frac{1}{2}$; (viii) $-\frac{1}{2}-x$, $-\frac{1}{2}-y$, $\frac{1}{2}-z$; (ix) -x, -y, -z; (ix) -x, -y, -z; (ix) -x, -x, -x; (iii) -x, -x, -x; (iii) -x, -x, -x; (iiii) -x, -x, -x; (iiii) -x, -x, -x; (iiii) -x, -x, -x; (iiii) -x, -x, -x, -x; (iiii) -x, -x, -x, -x; (iiii) -x, -x, -x, -x; (iii) -x, -x

The structure refinement of $ErCo_2Si_2$ clearly indicated that this phase crystallizes in the tetragonal crystal system in space group I4/mmm, adopting the $CeGa_2Al_2$ structure type. Refinement in space group P4/nmm ($CaBe_2Ge_2$ structure type) was less satisfactory and resulted in higher values of the R factors and atomic displacement factors. The highest maximum residual electron density is located at a distance of 0.73 Å from the Er atom, and the deepest hole is 1.80 Å from the same atom.

Data collection: *CrysAlis CCD* (Oxford Diffraction, 2003); cell refinement: *CrysAlis CCD*; data reduction: *CrysAlis RED* (Oxford Diffraction, 2004); program(s) used to solve structure: *SHELXS97* (Sheldrick, 1997); program(s) used to refine structure: *SHELXL97* (Sheldrick, 1997); molecular graphics: *DIAMOND* (Brandenburg, 1999); software used to prepare material for publication: *SHELXL97*.

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